

Nanowhiskers of $K_2Ti_6O_{13}$ as a promoter of photocatalysis in anatase mesocrystals

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Abstract

Here we describe the development of a new mesocrystalline anatase-based photocatalyst with an activity nearly 2 times exceeding the commercial Evonik Aeroxide[®] TiO₂ P25 catalyst. An extensive characterization of this new catalytic material with XRD, TEM, SAED, EELS, SEM, RAMAN and XPS shows that its activity is promoted by its hybrid structure comprising mesocrystalline anatase and $K_2Ti_6O_{13}$ whiskers formed within the TiO₂ mesocrystals. This new and unexpected component is formed *in situ* within the TiO₂ mesocrystalline framework at a surprisingly low temperature of 500 °C as a result of potassium diffusion from polymer-matrix

phase. The growth of whiskers occurs at the 001 TiO₂ planes of mesocrystals and their sizes attain 60 nm in a section and a few μm in length.

keywords: TiO₂ mesocrystals, composites, K₂Ti₆O₁₃ whiskers, hybrid photocatalyst

1. Introduction

Mesocrystals, the subject of a recent excellent review [1], may be viewed as macroscopic assemblies of crystallographically-organised nano-sized particles. Critically, the assembled superstructures retain the unique size-related properties of the individual nano-sized building blocks. This, coupled with the fact that their assembly processes are still not fully understood, means that mesocrystals are still the subject of significant research interest despite being first described in 2005 [2]. In the past decade mesocrystalline anatase has been extensively studied, primarily to better elucidate mesocrystal formation processes but also due to its ultimate photocatalytic activity, promoting various utilization from self-cleaning windows to water purification. Microcrystalline TiO₂ also holds a promise for various applications including electronic devices, medicine and catalysis. In a recent report we described the formation of anatase mesocrystals from crystalline precursor NH₄TiOF₃ and showed that the TiO₂ mesocrystals, so formed, exhibited modest photocatalytic activity [3]. Keen to explore and so better understand mesocrystal formation processes we set out to repeat this work and expand it by evaluating TiO₂ mesocrystal preparation under a variety of different reaction conditions. Herein we report how this study led to the development of a catalytic system that has better photocatalytic activity than Evonik Aeroxide[®] TiO₂ P25, a commercial titania-based photocatalyst with high catalytic activity comprising anatase (~81 %), rutile (~15 %) and amorphous (~4 %) titania [4]. Extensive characterisation of this new material showed it to be a composite material of TiO₂ mesocrystals bearing whisker-like protrusions of a second crystalline material, K₂Ti₆O₁₃. A number of publications have previously discussed the photocatalytic activity of sodium or potassium

hexatitanate in the photodegradation reactions of dyes methyl orange [5–7], rhodamine B [8], methylene blue [9], photoreduction of carbon dioxide to formaldehyde and methanol [10–12], photocatalytic water splitting [13,14]. Interestingly, Wu et al. [15] showed that a hybrid material, based on TiO_2 and $\text{K}_2\text{Ti}_4\text{O}_9$, had increased photocatalytic activity compared with pure samples of either titanium dioxide or potassium tetratitanate. Regarding synthesis, the formation of $\text{K}_2\text{Ti}_6\text{O}_{13}$ from TiO_2 directly has been reported to typically require reaction temperatures in excess of 800 °C [16–19]. For instance $\text{K}_2\text{Ti}_6\text{O}_{13}$ can be formed directly from TiO_2 and KF at 720 °C [20] or from TiO_2 via intermediate structures, such as $\text{K}_2\text{Ti}_4\text{O}_9$, at 1150 °C [21] or $\text{K}_2\text{Ti}_2\text{O}_5$ at 800–1100 °C [22], as a coating from Ti-foil by anodization in KOH [23], and by reaction of $\text{Na}_2\text{Ti}_3\text{O}_7$ in KOH at 250 °C for 3 days [24].

In this paper, a positive effect of the K-rich photoactive phase in titanium dioxide mesocrystal obtained at low temperatures is proposed.

2. Experimental

2.1. Synthesis

2.1.1. General method for the preparation of NH_4TiOF_3 MCs

$(\text{NH}_4)_2\text{TiF}_6$ (Sigma-Aldrich UK) 0.1 mol L⁻¹ poly(ethylene glycol) PEG-6000 (AlfaChemicals Ltd.) and gelation agent H_3BO_3 (Alfa Chemicals Ltd.) 0.2 mol L⁻¹ were dissolved in distilled water (30 mL) under continuous stirring. After full dissolution of the reagents, the resultant gel was kept at 35 °C for 20 hours. The resultant precipitate was isolated by centrifugation/decantation and subsequently washed with water (3 × 20 mL) and acetone (3 × 20 mL).

2.1.2. General Method for the Preparation of TiO_2 MCs

A sample of NH_4TiOF_3 MCs (ca. 0.5 g) was heated in air at 500 °C for the specified time period (2, 4 or 8 hours). This heating procedure was conducted using a Nabertherm HTCT 03/14

furnace and exhaust gases were vented into a designated fume cupboard (Caution: Gaseous HF is liberated in this thermally-mediated transformation).

2.2. Characterization

X-Ray diffraction (XRD) was conducted on a Bruker D8 Advance system using $\text{CuK}\alpha$ radiation. SEM was performed on a Carl Zeiss NVision 40 electron microscope. TEM was performed on a FEI Tecnai G2 F30 electron microscope with resolution of 0.14 nm. Raman spectroscopy was conducted using a Renishaw inVia Reflex spectrometer with an illumination wavelength of 633 nm. BET Low temperature nitrogen adsorption measurements were conducted using an ATX-6 analyzer (Katakon, Russia). Before measurements the samples (~300 mg weight) were outgassed at 200 °C for 30 min under a dry helium flow. Determination of the surface area was carried out by the 5-point Brunauer, Emmett and Teller (BET) method at the relative pressure range of $P/P_0 = 0.05\text{--}0.25$. X-ray photoelectron spectroscopy (XPS) analysis was performed on a SPECS laboratory spectrometer. To excite photoelectrons, magnesium anode X-ray radiation was used ($\text{Mg K}\alpha = 1253.6$ eV) with a tube voltage of 12 kV and emission current of 20 mA.

2.3. Photocatalytic test

To determine the photocatalytic activity of samples, we used a model reaction of methylene blue photodecomposition in an aqueous medium. For each sample photocatalytic activity was measured at room temperature. The photocatalyst (2 mg) was dispersed in 20 ml of aqueous solution of methylene blue dye (concentration of MB was 40 mg/L). Before illumination, the suspension with MB was stirred in the darkness for ~45 min to reach the adsorption/desorption equilibrium. The sample was excited in the wavelength range between 320 and 480 nm using the filtered output of a 300 W Xe lamp to provide a beam with 1.4 cm² excitation area and 30 mW/cm² excitation power. Aliquots (1 ml) were periodically withdrawn from the reaction mixture. Photocatalysts were separated by centrifugation at 8000 RPM for 10 minutes before UV-Vis analysis.

3. Results and Discussion

3.1. Strategy for the synthesis of the TiO₂-based mesocrystals photocatalyst

Mesocrystalline anatase is readily formed following extended heating (4-8 hours) of NH₄TiOF₃ at a temperature of or above 450 °C [3,25–29]. This process typically results in TiO₂ crystals that closely resemble the overall morphology of the precursor crystals of NH₄TiOF₃. This precursor is formed readily from the reaction between (NH₄)₂TiF₆ and H₃BO₃ in the presence of a polymer template/promoter such as PEG or BRIJ [30]. The effect of the concentration of PEG-6000 on the formation of NH₄TiOF₃ and TiO₂ mesocrystals was evaluated in our previous article [3]. Specifically, only a 1: 2: 4 molar ratio of PEG-6000: (NH₄)₂TiF₆: H₃BO₃, followed by heating yields TiO₂ mesocrystals. In the absence of PEG no mesocrystals form, a 1: 4: 8 ratio of reagents leads to the formation mesocrystals with a centralised defect containing nanoparticles of TiO₂ whilst a 3: 4: 8 molar ratio furnishes intergrown and aggregated mesocrystals. Accordingly, in this work the optimized 1: 2: 4 reagent ratio (see experimental section) was employed to generate precursor NH₄TiOF₃ mesocrystals and the only significant difference being that the final TiO₂ samples were to be obtained following heating at the slightly higher temperature of 500 °C for periods of 2, 4 and 8 hours.

XRD analysis showed that the two samples generated by heating for 2 and 4 hours respectively are single-phase anatase (fig. 1). In contrast, the sample heated for 8 hours is non-single-phase being a mixture of anatase and jeppeite phases (fig. 1). Raman spectroscopic data, obtained from single macroparticles, confirms that the 2- and 4-hour samples are pure anatase. Specifically, peaks at 144, 197, 399, 519 and 639 cm⁻¹ all correspond exactly to anatase [31]. The eight-hour sample, in addition to anatase peaks also shows second phase reflections as potassium hexatitanate (fig. 2). Two signals at ca 224 cm⁻¹ and 279 cm⁻¹ are consistent with the production of K-doped material [32] and an analogous Na-doped material [33]. Evidence for TiO₂ mesocrystal formation is provided by TEM and SAED (selected area electron diffraction) (fig. 3a,b). For example, material obtained by annealing at 500 degrees for 4 hours, gives single-crystal-like diffraction spots with minor distortion that are typical for mesocrystals (fig. 3b). The presence of

hexatitanate can be most clearly seen in SEM images (fig. 4d). Micrographs fig. 4b and fig. 4c show only anatase mesocrystals, which differ only in the size of their constituent nanocrystallites, while SEM image fig. 4d clearly shows a composite microparticle comprising a TiO₂ mesocrystal and nanowhiskers of K₂Ti₆O₁₃. The dimensions of the whiskers, only apparent in the sample heated for 8 hours, are roughly 60 nm across and between 1-2 μm in length. The evolution of crystallite size of TiO₂ nanoparticles was analyzed using both XRD and SEM data (Table 1). According to SEM data, an increase in the annealing time, results in crystallites growing in size from 30 to 90 nm. XRD data is consistent in showing an increase in size with increasing annealing time but, with this technique, a less significant increase from 45 to 53 nm is observed. Furthermore, it is notable that macroparticle thickness tends to increase with increasing annealing duration. Such a significant growth of crystallites according to the SEM data is most likely associated with the coalescence and growth of nanoparticles on the surface of the mesocrystal, whereas the XRD data relates to the sizes of crystallites within the entire volume of the mesocrystal. Thus, collectively our data and visualisations show, for the first time, the transition from non-classical to classical growth for TiO₂ mesocrystals. In summary, it appears that after 4 hours of annealing at 500 °C degree, TiO₂ mesocrystals become unstable and nanoparticles then begin to grow by a classical pathway [1] in all directions.

3.2. Characterization of the TiO₂-hybrid materials

To understand the root of formation of the composite material, a number of additional analytical techniques were employed. Firstly, XPS analysis (Table 2) confirmed the presence of potassium, likely a contaminant from the PEG [3], in the whiskers and also showed a significant reduction in fluorine had occurred in the final 4 to 8 hours of heating. XRD, TEM, SEM, RAMAN and XPS data collectively led us to conclude that, to our surprise, we had effected a thermally-mediated transformation from the parent mesocrystalline anatase into a hybrid TiO₂-K₂Ti₆O₁₃ material with two different crystal phases with very different morphologies being evident. As the annealing duration was increased the reduction of fluorine in the mesocrystals can be the cause of

instability of the inner pattern followed by lattice disturbance and nucleation of TiO_x-structures. The potassium present in the polymer-matrix PEG6000 most probably initiated the K₂Ti₆O₁₃ formation.

According to the TEM, SAED (fig. 5) and EELS data (Table 3), we observe that the nanowhiskers are single-phase potassium hexatitanate. The potassium content is the same along the length of the nanowhisker (fig. 6 and Table 3). We assume that nanowhisker growth occurs as follows - potassium ions diffuse to particles of fluorine-doped TiO₂ and the growth of K₂Ti₆O₁₃ occurs according to the classical mechanism [34].

3.3 Photocatalytic performance

The presence of potassium hexatitanate in the hybrid highly oriented material prompted us to anticipate that there would be an increase in photocatalytic activity as compared with anatase mesocrystals. To test this hypothesis, a standard photocatalytic assay procedure employing a singlet oxygen mediated-bleaching of methylene blue was conducted on composite samples generated by heating at 500 °C for 2, 4 and 8 hours.

Before the photodecomposition test, concentration change of dye before and after adsorption/desorption equilibrium was measured. There was no obvious decolorizing observed. Although P25 has higher specific surface area and thus larger amount of dye absorption was expected, due to small amount of photocatalysts presented in suspension, the amount of dye molecules were highly excessive to the surface area (i.e. surface adsorption sites) of any tested photocatalyst, the difference in absorbed dyes in various photocatalysts was negligible in this test condition.

The photodecomposition results obtained from this study are shown in fig. 7 and show that increased heating times generates material with increased photocatalytic activity. Moreover, the sample formed after heating for 8 hours showed excellent photocatalytic activity that is significantly greater than that of the commercial Evonik Aeroxide[®] TiO₂ P25 catalyst. This last observation was most surprising since in our previous work [3] we had observed only relatively

modest photocatalytic activities which increased as the sample heating time increased from 2 to 4 hours at 450 °C but which decreased again after extended heating (i.e. 8 hours). This loss of activity coincided with a loss of mesocrystalline structural integrity and a concomitant reduction in surface area as assessed by BET. Interestingly, for samples obtained by annealing at 500 °C, the specific surface area does not undergo significant change (Table 1). The probable decrease in the specific surface area associated with an increase in the size of nanocrystallites in the mesocrystal during an increase in the duration of annealing is most likely compensated for by the formation of nanowhiskers.

Analysis of the dye decomposition data revealed that the reactions follow first-order kinetics (fig. 8). The apparent reaction rate constant k of mesocrystal TiO₂ samples increased from 6.04×10^{-3} to $1.74 \times 10^{-2} \text{ min}^{-1}$ when the duration of annealing increased from 2 to 8 hours. The 8 hours annealed TiO₂-K₂Ti₆O₁₃ hybrid showed 86% higher photocatalytic activity than the commercial photocatalyst Evonik Aeroxide® TiO₂ P25. In contrast, such an increase in annealing time at 450 °C more than halved the photocatalytic activity [3].

The mechanism of potassium titanate influence definitely worth further investigation. In particular, one dimension and high crystalline nature of K₂Ti₆O₁₃ nanowhisker and heterojunction structured at the interface between K₂Ti₆O₁₃ and TiO₂ are being looked into. Both features can enhance photogenerated charge transfer and separation, and thus contribute to promoted photocatalytic activities. Further study will follow to reveal more insights.

4. Conclusions

A hybrid material of anatase mesocrystals and potassium titanate (K₂Ti₆O₁₃) nanowhiskers with high photocatalytic activity is successfully formed from ammonium oxofluorotitanate mesocrystals in presence of PEG-6000. For the first time, the phase of potassium hexatitanate from ammonium oxofluorotitanate is achieved by solid-state synthesis at a low temperature of 500 °C. The results of SEM, TEM and SAED show that the growth of potassium titanate (K₂Ti₆O₁₃) nanowhiskers occurs oriented to the 001 TiO₂ plane of the mesocrystals. The obtained material

demonstrates a photocatalytic activity nearly 2 times higher than that of a standard commercial photocatalyst, Evonik Aeroxide[®] TiO₂ P25, for the photodegradation reaction of methylene blue. Thus, the methodology described herein opens up the possibility of obtaining TiO₂-K₂Ti₆O₁₃ hybrid materials with high crystallinity and excellent photocatalytic activity for application in, for example, water splitting reactions.

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Table 1. Parameters of NH_4TiOF_3 and TiO_2 materials

	BET, m^2/g	Planar size, μm	Thickness, μm	Nanoparticles size by XRD, nm	Nanoparticles size by SEM, nm
NH_4TiOF_3	1	2.0-2.5	0.8-1.1		-
meso TiO_2 after 2 h	6	2.3-2.9	1.2-1.4	45 ± 1	30+/-
meso TiO_2 after 4 h	11	2.6-3.2	1.6-1.8	47 ± 1	40+/-
meso TiO_2 after 8 h	10	3.1-3.8	1.7-2.0	53 ± 1	90+/-

Table 2. XPS for TiO_2 obtained after 2, 4 and 8 hours at 500 °C.

	O	Ti	K	F	Ti/O	K/Ti	F/Ti
meso TiO_2 after 2 h	50.6	21.1	14.8	13.4	0.41	0.70	0.63
meso TiO_2 after 4 h	52.7	20.3	14.2	12.8	0.38	0.69	0.63
meso TiO_2 after 8 h	60.4	23.0	14.0	2.5	0.444	0.61	0.10

Table 3. EELS for $\text{K}_2\text{Ti}_6\text{O}_{13}$ obtained after 8 hours at 500 °C.

	K	Ti	K/Ti
Spectrum 1	18.7	81.3	0.23
Spectrum 2	20.2	79.8	0.25
Spectrum 3	18.3	81.8	0.22
Spectrum 4	20.1	79.9	0.25

FIGURE CAPTIONS

Fig. 1. Wide-angle (A) and small-angle (B) powder XRD patterns of hybrid materials based on TiO₂-anatase after anneal 2, 4 and 8 hours

Fig. 2. Raman spectra of hybrid materials obtained after annealing for 2, 4 and 8 hours at 500 °C.

Fig. 3. HRTEM (A) and SAED pattern (B) images for the TiO₂ mesocrystal obtained after annealing for 4 hours at 500 °C

Fig. 4. SEM images of NH₄TiOF₃ (A) and hybrid materials based on TiO₂-anatase after annealing for 2 (B), 4 (C) and 8 (D) hours at 500 °C

Fig. 5. HRTEM and SAED pattern images for the K₂Ti₆O₁₃ nanowhiskers

Fig. 6. TEM image of a single K₂Ti₆O₁₃ whisker obtained after heating ammonium oxofluorotitanate for 8 hours at 500 °C

Fig. 7. Methylene blue dye photodecomposition rates in the absence and presence of various samples of titanium dioxide (after annealing for 2, 4 and 8 hours at 500 °C) and reference material - Evonik Aeroxide® TiO₂ P25

Fig. 8. First-order reaction kinetics plots of methylene blue dye photodecomposition rates.

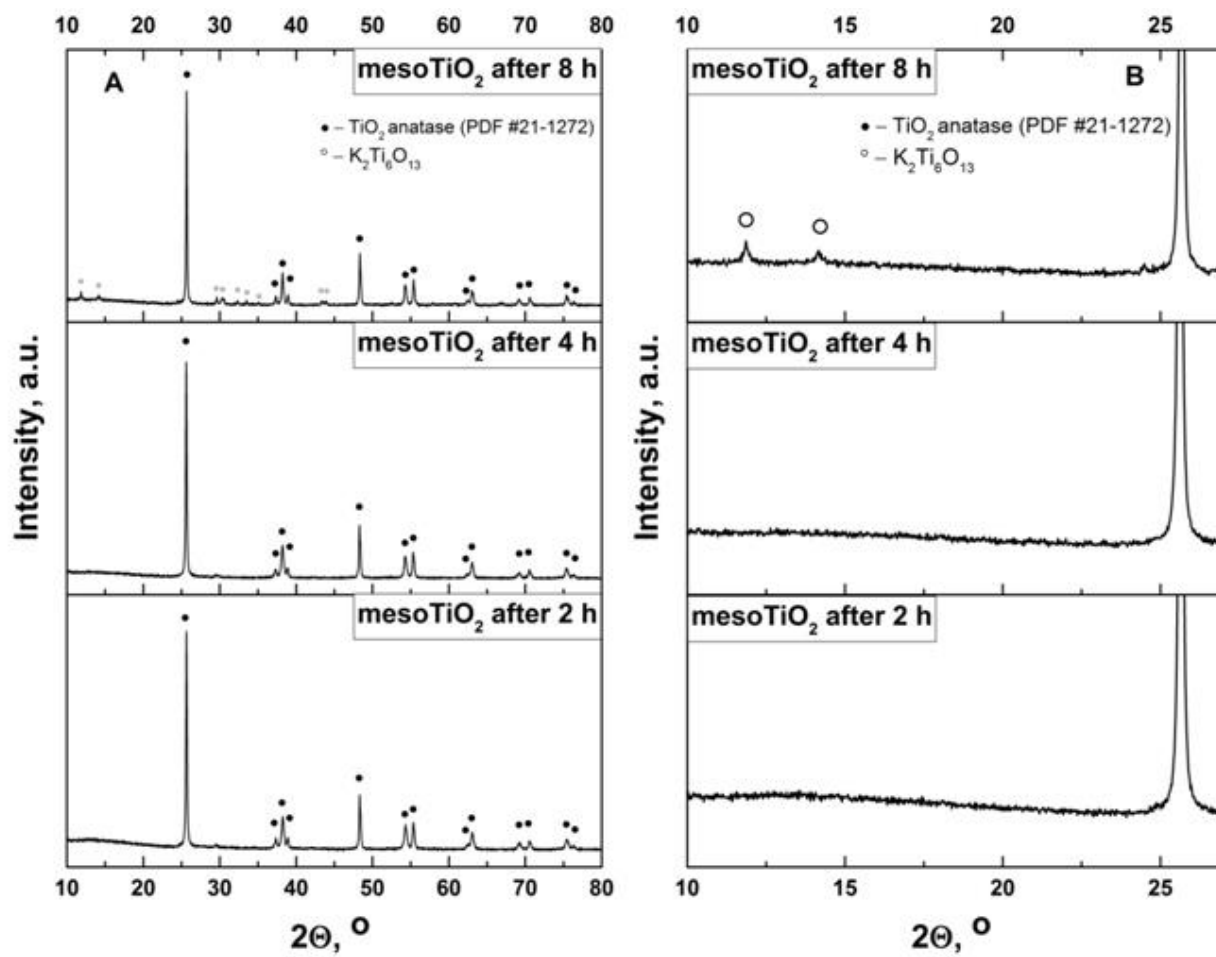


Fig. 1.

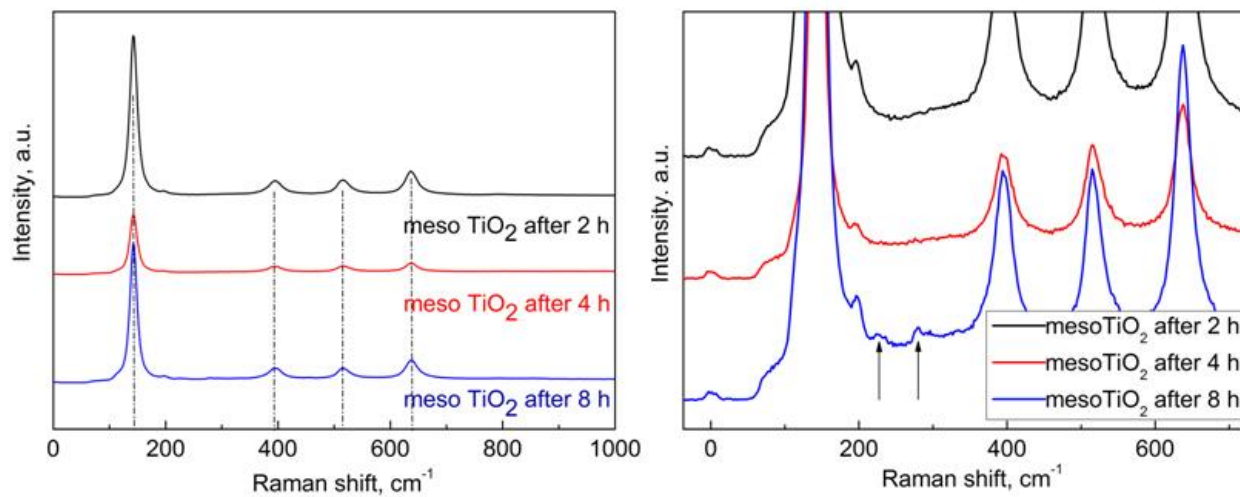


Fig. 2.

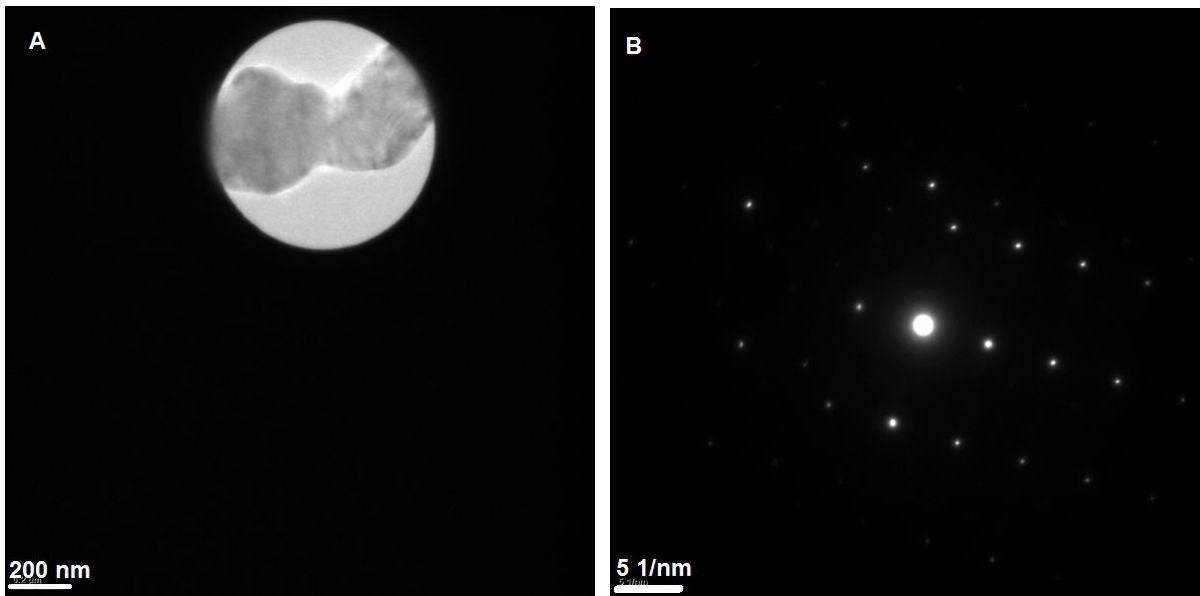


Fig. 3.

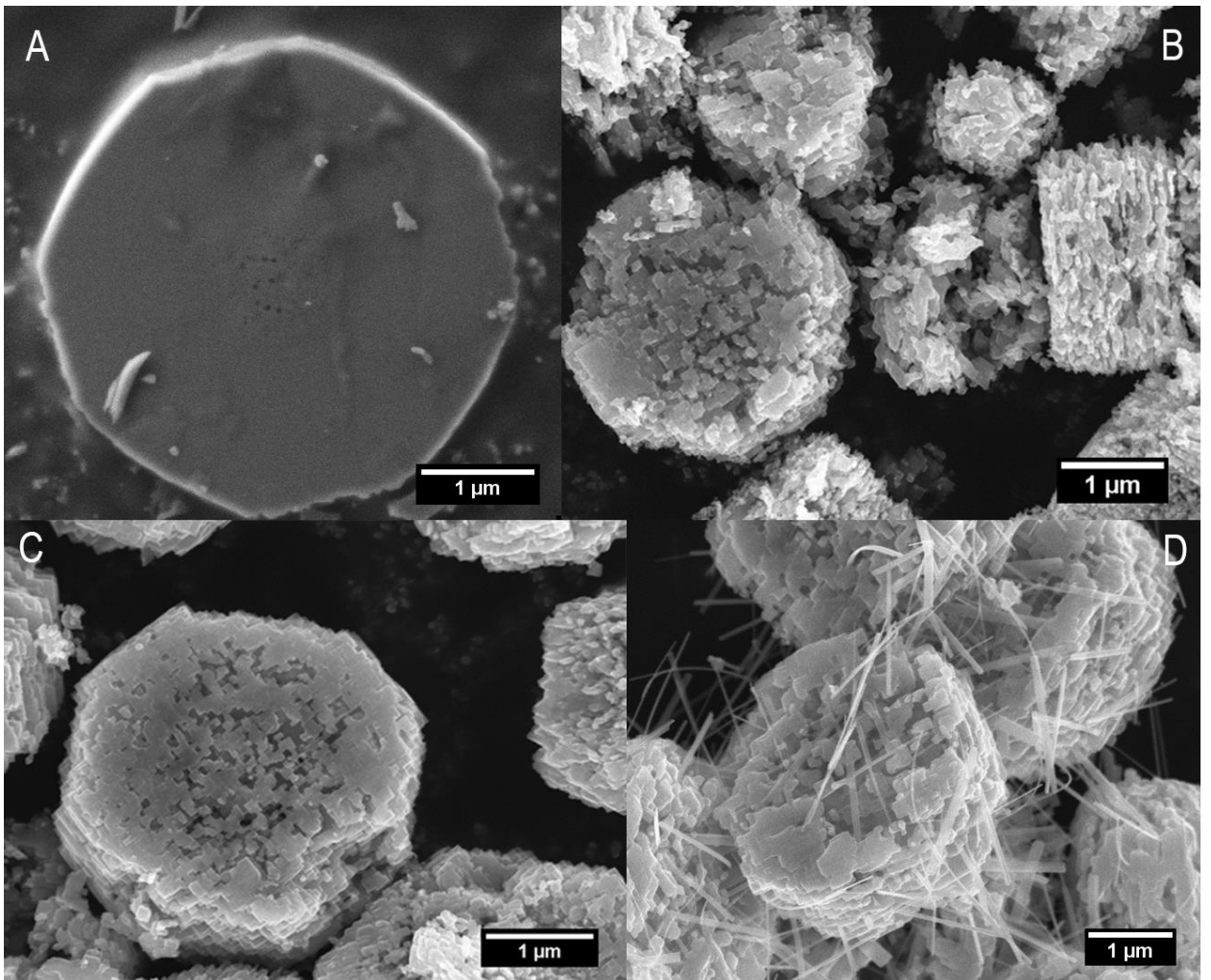


Fig. 4.

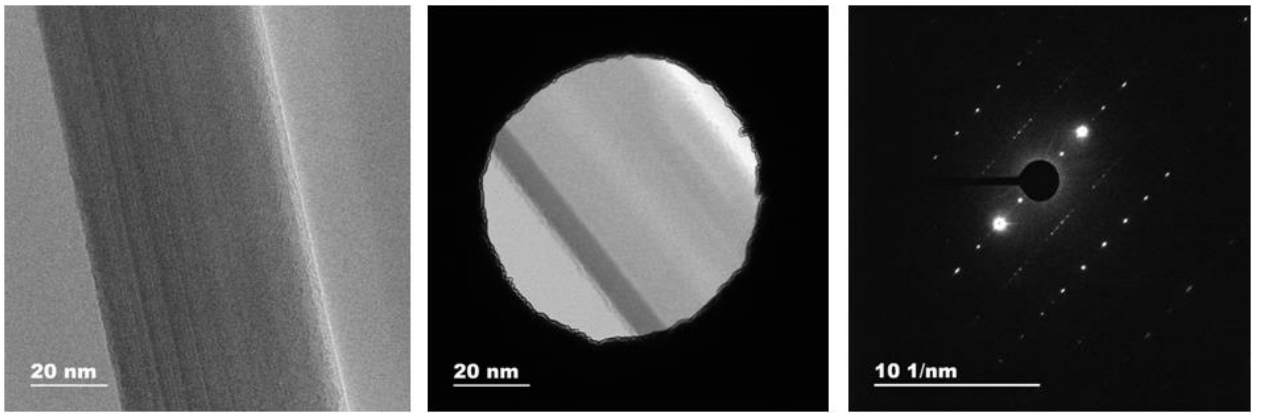


Fig. 5.

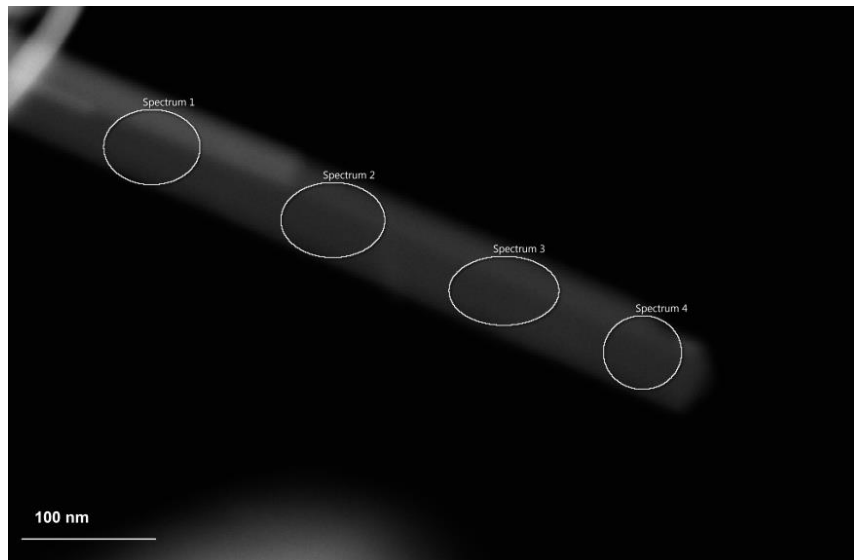


Fig. 6.

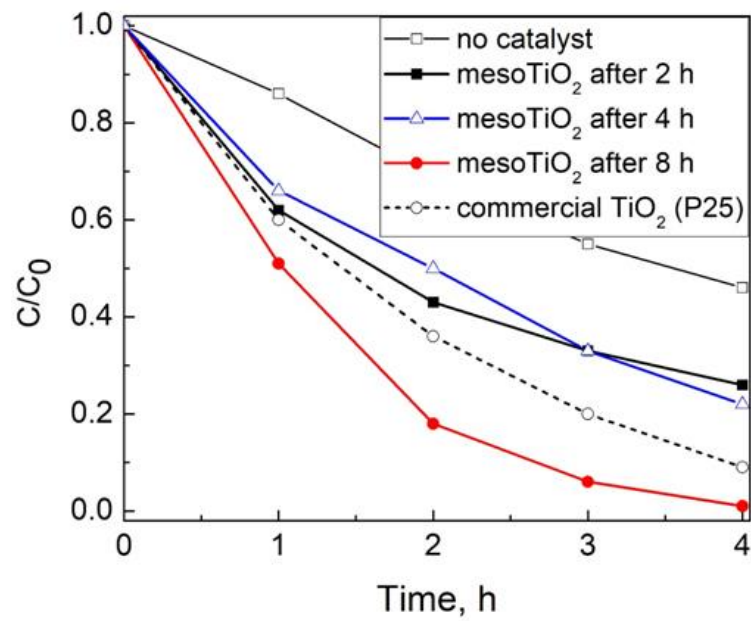


Fig. 7.

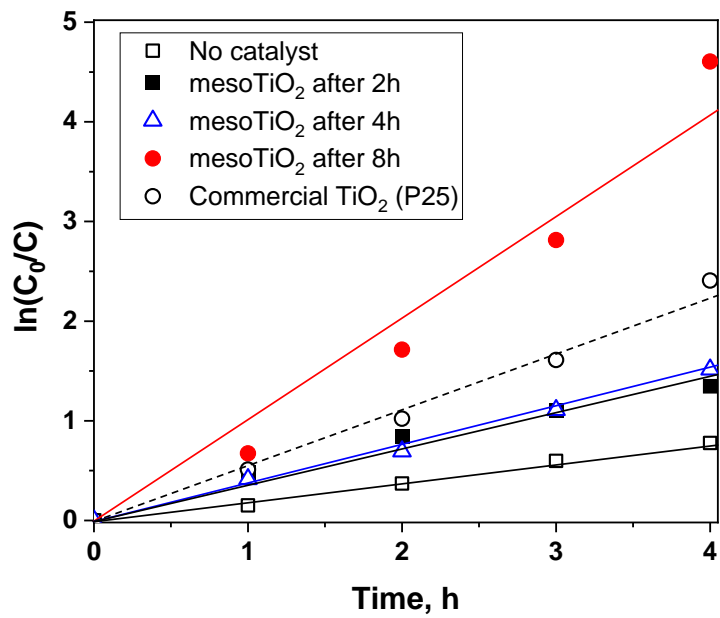


Fig. 8